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# Heterojunction nanoarchitectonics of $WO_x/Au-g-C_3N_4$ with efficient photogenerated carrier separation and transfer toward improved NO and benzene conversion

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# ABSTRACT

Topics on effectively improving the photochemical CO<sub>2</sub>/benzene/NO oxidation conversion performances of g-C<sub>3</sub>N<sub>4</sub> based materials via charge transfer and separation enhancement are still considered challenging, despite the growing popularity of applying these materials in a variety of energy conversion related applications. Based on the idea of nanoarchitectonics, a post-nanotechnology concept, WO<sub>x</sub>/Au-g-C<sub>3</sub>N<sub>4</sub> heterostructures are synthesized using two-step thermal polymerization and solvothermal treatment methods in this paper. Small Au nanoparticles are incorporated in superior thin g-C<sub>3</sub>N<sub>4</sub> via mechano-chemical pre-reaction and two-step thermal polymerization (treated at 500 and 700 °C). Enhanced photocurrent density is observed after incorporation of Au, which is also in good agreement with the photocatalytic activity (H<sub>2</sub> generation and  $CO_2$  reduction) data. Layered  $WO_x$  with abundant oxygen vacancies are further incorporated into Au-modified g-C<sub>3</sub>N<sub>4</sub> nanosheets to form heterojunctions possessing excellent photocatalytic CO<sub>2</sub> photo-reduction performances with CO and CH<sub>4</sub> generation rate of 5.64 and 2.58  $\mu$ molg<sup>-1</sup>h<sup>-1</sup>, respectively, under full solar spectrum. The heterojunctions constructed via in-situ formation show direct Z-scheme charge transfer pathway with improved charge separation and transport efficiencies. These highly stable and recyclable hierarchical g-C<sub>3</sub>N<sub>4</sub> hybrid nanostructures  $(WO_x/Au-g-C_3N_4$  heterojunctions) show outstanding conversion rate (88.1%) and selectivity (99.3%) for benzene to phenol conversion under full solar spectrum condition, as well as excellent NO removal rate (61%).

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#### 1. Introduction

Fossil fuel (involving coal, petroleum, heavy oils, natural gas etc.) combustion has been considered the primary cause for rising  $CO_2$  and  $NO_x$  levels in the atmosphere that leads to anthropogenic climate change [1,2].  $CO_2$  has been generally believed to be one of the major contributors to the global warming by supercharging natural greenhouse effect; while  $NO_x$  emission been considered to be responsible for anthropogenic smog and acid rain formation. Techniques such as photocatalytic  $CO_2$  and  $NO_x$  conversion, are promising choices that reduces atmospheric  $CO_2$  and  $NO_x$  levels whilst reducing fossil fuel reliance in a sustainable way [3–5].

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Various catalysis techniques involving electrocatalysis, photocatalysis, photoelectro-catalysis, and thermocatalysis, have been developed for converting  $CO_2$  into valuable chemical fuels (e.g., methane, formaldehyde, methanol and other carbon-based fuels) [6–8]. During the conversion process, clean and sustainable energy source with high abundance (e.g., solar energy) as well as highefficiency catalysts are often required to activate the quite stable  $CO_2$  molecules, photocatalytic activation of  $CO_2$  has been considered high potential choice [9–11].

Inspired by the natural photosynthesis, photocatalysis has been considered a promising energy conversion technology for converting  $CO_2$  into solar fuels (e.g., carbon-based fuels), and  $H_2O$  is known to play a dominant role in these  $CO_2$  photoreduction processes (involving light absorption step, charge separation process, and redox reaction process) [12]. Topics on the photocatalytic conversion of  $CO_2$  and  $H_2O$  into carbon-based fuels using semiconductor catalysts have been widely studied [13–15]. Among a

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variety of semiconducting photocatalysts, graphitic carbon nitride (g-C<sub>3</sub>N<sub>4</sub>) has been drawn considerable attention in photocatalytic solar energy conversion due to its high stability, nontoxicity and visible (Vis) light responsiveness [16–18]. Since g-C<sub>3</sub>N<sub>4</sub> nanosheets are disadvantaged by its limited visible light absorption and high photogenerated charge carrier recombination rate in CO<sub>2</sub> conversion process, development of g-C<sub>3</sub>N<sub>4</sub> based heterostructures with enhanced photocatalytic performances has become a quite popular topic [19–21]. The ordered unit arrangement and interlayer structure of C<sub>3</sub>N<sub>4</sub> are often formed through pyrolysis of layered supramolecular precursors. The supramolecular layered precursors possess abundant functional groups (e.g., -NH<sub>2</sub> and -OH in between heptazine layers) and large interlamellar distance that allows facile intercalation of small guest species (e.g., metal salts) forming heterostructures [11]. And tremendous efforts have been done on the optimization of these heterostructures to improve their photocatalytic performances for practical applications.

WO<sub>3</sub> having a band gap of around 2.6–2.9 eV has also been considered a high-performance visible-light-responsive photocatalyst [22-25]. However, the relatively lower conduction band (CB) edge position of WO<sub>3</sub> photocatalyst often leads to low reactivity towards photocatalytic CO<sub>2</sub> reduction and NO oxidation reactions as well as other photo-reduction/oxidation reactions. In order to improve the photogenerated charge carrier separation efficiency as well as the photocatalytic and electrochemical properties of WO<sub>3</sub> incorporated catalyst, oxygen vacancies can be introduced into WO<sub>3</sub> by defect engineering [26,27]. For example, specific cooling methods were utilized by Zhang et al. to introduce oxygen vacancies on exposed specific facets of WO<sub>3</sub> to attain improved photocatalytic oxidation activity [28]. Among a variety of WO<sub>3</sub> incorporated composites developed, two dimensional (2D)/ 2D heterostructures with the advantages of enlarged interfacial contact surface and improved charge carrier migration efficiency have been considered one of the most competitive choices. Although g-C<sub>3</sub>N<sub>4</sub> based WO<sub>3</sub> 2D/2D heterostructures have been widely studied for various photocatalytic applications (involving organic pollutants treatment, photocatalytic water splitting and  $CO_2$  reduction etc.), development of high-performance g- $C_3N_4/WO_3$ based heterostructure catalysts still remains challenging.

For Z-scheme heterostructural system with multi-components, transport of photoexcited electrons occurs from the conduction band of catalyst component A to the valence band of catalyst component B, with photogenerated electrons and holes accumulated at the conduction band of component B and the valence band of component A, respectively [26,29,30]. And these photogenerated charge carriers often possess high reduction/oxidation capacity for photocatalytic reduction and oxidation in water splitting as well as CO<sub>2</sub> and NO<sub>x</sub> conversion processes. In order to improve the charge transport efficiency in Z-scheme heterostructural system, deposition of metal nanoparticles has been an effective approach. The metal particles work as the electron acceptor for receiving electrons from the conduction band of the semiconductor photocatalyst. Among a variety of popular metal choices, gold as a plasmonic noble metal with surface plasmon resonance (SPR) effect, is the key for the photocatalytic activity enhancement of the heterostructure composites. The theory behind is that the enhanced localized electromagnetic fields and the SPR-induced plasmonic hot electrons are responsible for the enhanced interfacial charge carrier separation at the close contact of metal and semiconductor [31–33]. And by eliminating the potential interfacial charge transport barrier commonly encountered in type II heterostructure composites via Z-scheme heterostructure construction, improvement in charge separation efficiency and photocatalytic activity of these Au nanoparticles incorporated g-C<sub>3</sub>N<sub>4</sub> based Z-scheme composite catalyst can be achieved.

Attention has been paid to the photocatalytic removal of NO (one of the most hazardous constituents in the  $NO_x$  gases family) and benzene-to-phenol conversion utilizing semiconducting catalysts in recent years, due to the high potentials of semiconductor catalysts in photocatalytic energy conversion and pollutants treatment applications [31-34]. Hydroxylated aromatic compounds (e.g., phenol) are often used as precursors or intermediates for the fabrication of plastic resins, dves, and fibers in commercial and industrial scale. Phenol production in industrial scale often involves partial oxidation of isopropyl benzene via the high-energy consuming cumene process with acetone produced as the byproduct accompanied by an extremely low phenol yield of about 5%. It is thus imperative to develop more environmental-friendly alternative methodologies for benzene production. Solar energy conversion utilizing low-loading noble metal (Au in this case) modified carbon-based photocatalysts has been given special attention, because of the extremely high catalytic activity of Au and also due to the fact that the high chemical/thermal stability as well as the large surface area makes layered g-C<sub>3</sub>N<sub>4</sub> a compelling base material for visible-light-driven benzene/NO<sub>x</sub> oxidation [33].

As a trending post-nanotechnology concept, nanoarchitectonics (first introduced by Aono and co-workers in 2000) based on the idea of construction of multifunctional nanomaterials via combining/coupling different functional nanomaterials, has been drawing growing attention in nanofabrication recent years [35-38]. In this paper, a WO<sub>x</sub> and Au modified g-C<sub>3</sub>N<sub>4</sub> heterostructure having enhanced photochemical properties was constructed using the idea of nanoarchitectonics. Mechano-chemical treatment and thermal condensation process (on bulk  $g-C_3N_4$ precursor) plays the key role in successful incorporation of homogeneously distributed small Au nanoparticles (NPs) and Au clusters in ultrathin g-C<sub>3</sub>N<sub>4</sub> nanosheets. A 2D/2D WO<sub>x</sub>/Au/g-C<sub>3</sub>N<sub>4</sub> Z-scheme heterostructure was formed via in situ growth of WO<sub>x</sub> nanobelts on Au modified g-C<sub>3</sub>N<sub>4</sub> using solvothermal treatment method. Characterization techniques such as transmission electron microscopy (TEM), X-ray photoelectron spectroscopy (XPS), and Raman spectroscopy were applied to investigate the composition, microstructures, and photocatalytic mechanism of these heterostructures. Excellent photocatalytic CO<sub>2</sub> reduction performances (5.64  $\mu$ molg<sup>-1</sup>h<sup>-1</sup> for CO and 2.58  $\mu$ molg<sup>-1</sup>h<sup>-1</sup> for CH<sub>4</sub> generation) were observed for the WO<sub>x</sub>/Au/g-C<sub>3</sub>N<sub>4</sub> Z-scheme heterostructures. In addition, outstanding conversion rate (88.1%) and selectivity (99.3%) for benzene to phenol conversion under full solar spectrum condition, as well as excellent NO removal rate (61%) were created using these heterostructures.

# 2. Experimental section

*Chemicals and preparation of Au/g-C<sub>3</sub>N<sub>4</sub>*. Chemicals involving chloroauric acid (AuClO<sub>3</sub>, 97%, as Au source), ascorbic acid (AA), tungsten hexachloride (WCl<sub>6</sub>), melamine, Nafion solution, rhodamine (RhB), and ethanol were purchased from Sigma Aldrich and used with no further purification. The pristine ultrathin g-C<sub>3</sub>N<sub>4</sub> nanosheets (sample CN) with no Au incorporation (for comparison uses) was synthesized via a two-step calcination process using one of our previously reported methodologies [16-18]. Mechanochemical pre-treatment and multi-step thermal polymerization process (at high temperature settings) were used to fabricate Au modified g-C<sub>3</sub>N<sub>4</sub> (Au–CN). Bulk g-C<sub>3</sub>N<sub>4</sub> (as the precursor for Au–CN composite sample synthesis) was prepared via thermal polymerization of melamine at 500 °C. Certain amount of AuClO3 were added to the bulk g-C<sub>3</sub>N<sub>4</sub> base to reach the weight percentage of 0.5, 1.0, and 1.5 wt% for sample Au-CN-1, Au-CN-2, and Au-CN-3, respectively (as shown in Table 1). These precursor mixtures were ground with ethanol to dry before undergoing calcination process

Sample	Composition	AuClO <sub>3</sub> (wt%) <sup>a</sup>	WCl <sub>6</sub> (wt %) <sup>a</sup>	Procedure
CN	g-C <sub>3</sub> N <sub>4</sub>	0	0	Two-step synthesis
Au-CN-1	$Au/g-C_3N_4$	0.5	0	Pre-reaction
Au-CN-2	$Au/g-C_3N_4$	1.0	0	Pre-reaction
Au–CN-3	$Au/g-C_3N_4$	1.5	0	Pre-reaction
W-Au-CN-1	WO <sub>x</sub> /Au/g-C <sub>3</sub> N <sub>4</sub>	1.0	6.0	Pre-reaction
W-Au-CN-2	WO <sub>x</sub> /Au/g-C <sub>3</sub> N <sub>4</sub>	1.0	8.0	Pre-reaction
W-Au-CN-3	$WO_x/Au/g-C_3N_4$	1.0	10.0	No pre-reaction

<sup>a</sup> The weight ratio of AuClO<sub>3</sub> and WCl<sub>6</sub> to g-C<sub>3</sub>N<sub>4</sub> used for sample preparation.

(in inert gas environment) in tube furnace (at 700  $^{\circ}$ C for 2 h, with a ramp rate of 5  $^{\circ}$ C/min) to obtain sample Au–CN-1, Au–CN-2, and Au–CN-3.

**Preparation of WO<sub>x</sub>/Au-g-C<sub>3</sub>N<sub>4</sub> heterojunctions.** WO<sub>x</sub> modified Au/g-C<sub>3</sub>N<sub>4</sub> heterojunctions were constructed using sample Au–CN-2 as the base precursor. To investigate the effects of WCl<sub>6</sub> (W source) on heterojunction formation, different amount of WCl<sub>6</sub> were mixed with Au/CN-2 to reach a weight percentage of 6.0, 8.0, and 10 wt% (for sample W–Au–CN-1, W–Au–CN-2, and W–Au–CN-3, respectively). 30 mL of ethanol was added to these powder mixtures followed by 5 min of sonication treatment. Scarce amount of AA was added to the obtained mixture (with a WCl<sub>6</sub> to AA molar ratio of 1:2 and constant stirring) followed by 20 min of stirring. The obtained suspension was then heat-treated in an oven at 220 °C for 10 h and washed with ethanol.

**Material characterization.** Transmission electron microscope (TEM, FEI Titan G2 80–200 TEM/STEM, FEI company) was used to collect the TEM and HAADF-STEM images of the Au–CN and W–Au–CN heterostructure samples. X-ray diffraction (XRD) patterns and the XPS spectra of the heterostructure samples were recorded using Bruker D8 X-ray diffractometer and Kratos Axis Ultra DLD spectrometer, respectively. UV–Vis diffuse reflectance and absorbance measurements were done on the heterostructure samples utilizing UV–Vis spectrophotometer (U-4100, Hitachi, Japan).

**Photocatalytic CO<sub>2</sub> reduction**. A mixture solution containing 20 mg of the heterostructure sample and 15 mL of deionized water

with the NO removal rate of the sample calculated using the following equation:  $\left(1 - \frac{C}{C_0}\right) \times 100\%$ . Cyclic NO oxidation stability test was also performed on the heterostructure samples in full solar spectrum condition having the light source (Xe lamp) turned off at the end of each cycle and with the duration of each cycle been about 8 min. In order to examine the active species in the photocatalytic NO oxidation reactions, quenching experiments were performed on the heterostructure samples. 2 mmol of the scavengers (benzoquinone (BQ), AgNO<sub>3</sub>, isopropanol (IPA), and ethylenediaminetetraacetic acid disodium salt (EDTA-Na<sub>2</sub>)) and 0.1 g of the catalyst sample were dispersed in 5 mL of ethanol to obtain a mixture solution followed by 5 min of sonication. The obtained suspension was then coated and dried on quartz glass sheets before testing.

**Photocatalytic oxidation of benzene.** A catalyst suspension containing 10 mg of the catalyst sample, 1 mL of benzene, 5 mL of CH<sub>3</sub>CN (as solvent) and 2 mL of 25%  $H_2O_2$  was prepared and placed 10 cm from the light source. The benzene hydroxylation test was performed on the samples in a sealed environment under full solar spectrum irradiation for 12 h (at 50 °C). Gas chromatography mass spectrometry (GC-MS) and GC were used for collecting and analyzing the data. Phenol yield was calculated by dividing the phenol generated (mole) by the initial mass of benzene (mole), and the selectivity to phenol was calculated using the equation below.

# Photocatalytic hydrogen generation and RhB degradation. A

 $Selectivity \ to \ Phenol = \frac{phenol \ generated \ (mole)}{phenol + benzoquinone \ generated \ (mole)} \times 100\%$ 

was prepared for  $CO_2$  photoreduction measurement. 20 min of high-purity  $CO_2$  bubbling was done on the solution before vacuum treatment and conducting measurements under full solar spectrum irradiation (utilizing a LED light source). A GC-7920 gas chromatograph was used for recording and analyzing the data collected at certain time intervals.

**Photocatalytic NO removal.** A device having a cylindrical reaction chamber connected to a  $NO_x$  analyzer and a NO compression cylinder was used for photocatalytic NO oxidation measurements (at room temperature with a NO gas (600 ppb) flow rate of 1.2 mL/min). A quartz glass sheet with 0.1 g of heterostructure sample was placed onto the bottom plate of the cylindrical reaction chamber followed by exposing to 25 min of NO gas flow under no light irradiation condition (in order to reach the NO adsorption-desorption equilibrium with an initial NO concentration of  $C_0$ ). 20 min of measurements were done on each sample under full solar spectrum irradiation condition (500 W Xe lamp as the light source)

testing system having glass-closed gas circulation and Pyrex topirradiation reaction vessel was used for photocatalytic hydrogen generation measurements. 25 min of sonication was done on the sample mixture containing 10 mg of the sample and 100 mL of triethanolamine solution (with a triethanolamine: deionized water volume ratio of 1:9) before removing air from the testing system. H<sub>2</sub> generation measurements were performed with constant stirring utilizing 300 W Xe lamp as the light source and argon as the Carrier gas. Shimadzu GC-7920 gas chromatography was used for data collection.

RhB photodegradation measurements utilizing 10 mg/L RhB solution were done on the heterostructure samples. A wavelength cut-off filter equipped 300 W Xe arc lamp was used as the light source. The active species in the RhB photodegradation process in full solar spectrum irradiation condition were investigated by adding scavengers in the catalyst sample solution (e.g., 0.0216 g of BQ, 0.034 g of AgNO<sub>3</sub>, 2 mL of IPA, and 0.0744 g of EDTA-Na<sub>2</sub> were

mixed with 35 mL of mixture solution containing catalyst sample and RhB solution). A Bruker model ER200-SRC spectrometer was used for recording the ESR signals of the superoxide and hydroxyl radical adducts of 5,5-dimethyl-1-pyrroline *N*-oxide (DMPO).

## 3. Results and discussion

Scheme 1 shows the preparation procedure utilized on synthesizing WO<sub>x</sub>/Au-g-C<sub>3</sub>N<sub>4</sub> heterostructure samples based on the concept of nanoarchitectonics. Melamine was used as the precursor to synthesize bulk layered g-C<sub>3</sub>N<sub>4</sub> consisted of melem building blocks (involving tri-s-triazine ( $C_6N_7$ ) and triazine ( $C_3N_3$ ) units) via thermal condensation process at 500 °C [39]. These layered g-C<sub>3</sub>N<sub>4</sub> has  $\pi$ -conjugated system containing sp<sup>2</sup> hybridized carbon and nitrogen. A mechano-chemical pre-reaction treatment was required for mixing bulk g-C<sub>3</sub>N<sub>4</sub> and the Au sources. The presence of abundant N sites in the g-C<sub>3</sub>N<sub>4</sub> nanosheets is able to promote the incorporation of Au ions on g-C<sub>3</sub>N<sub>4</sub> forming reaction sites so that Au can be anchored into the composite system during thermal polymerization process [24]. The Au nucleation and growth were triggered by the reduction of Au ions during the multi-step thermal polymerization process at 700 °C, leading to successful incorporation of Au clusters in superior thin g-C<sub>3</sub>N<sub>4</sub> nanosheets substrate. Solvothermal treatment method was used to achieve in situ growth of WO<sub>x</sub> nanobelts on the Au modified g-C<sub>3</sub>N<sub>4</sub> base at 200 °C. The detailed preparation conditions used for synthesizing each sample are illustrated in Table 1.

Fig. 1a shows the scanning electron microscopy (SEM) image of pristine  $g-C_3N_4$  with nanosheets morphology obtained via twostep thermal polymerization, while Fig. 1b to f shows the TEM images of sample Au–CN-2 (Au modified  $g-C_3N_4$  prepared using a metal source weight percentage of 1%) at different magnifications. No obvious chunky Au NPs were observed for sample Au–CN-2, and the high-resolution TEM images of the Au clusters area of Au–CN-2 were recorded with the metal clusters circled in Fig. 1d to f. Moreover, the nanosheets morphology of  $g-C_3N_4$  was maintained after Au loading, however the crystallinity of the  $g-C_3N_4$  nanosheets was enhanced, and this can be attributed to the multi-step thermal polymerization process utilized for Au cluster incorporation. The observed lattice fringes with a spacing of 0.32 nm corresponding to the (002) facet of  $g-C_3N_4$  nanosheets.

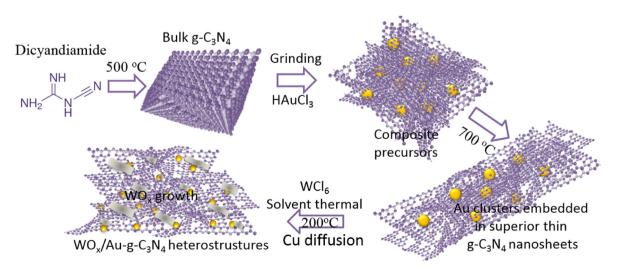
Fig. 1g to j showed the high-angle annular dark-field scanning TEM (HAADF-STEM) images and elemental mapping (of N, C, and

Au) for sample Au–CN-2. Homogeneously distributed Au with no obvious chunky Au NPs were observed for sample Au–CN-2, which also further confirms the successful embedment of Au clusters in Au–CN-2. However, as the weight percentage of metal source increased to 1.5% (sample Au–CN-3), small Au NPs with diameters of less than 6 nm were observed on the TEM images due to the increased Au loading (as shown in Fig. 1k, m, and 1n). According to literature, the lattice spacing of 0.23 nm observed on Fig. 1n corresponds to the (111) facet of Au [40].

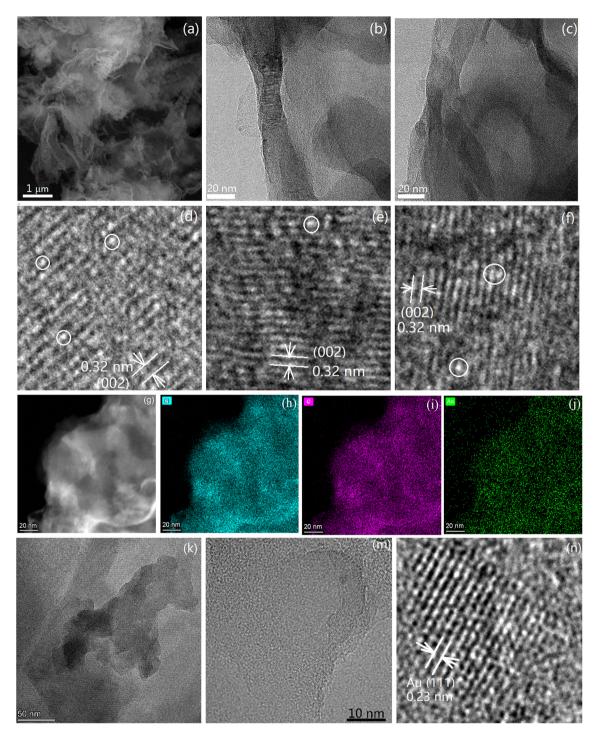
The WO<sub>x</sub> and Au modified sample W–Au–CN-1 still showed nanosheets morphology as shown in the SEM image of W–Au–CN-1 in Fig. 2a. Fig. 2b to d shows the TEM images of sample W–Au–CN-1 (at different magnifications) with limited amount of layered WO<sub>x</sub> nanobelts and no obvious chunky Au NPs observed on Fig. 2b and c. Some of the Au clusters (incorporated in the W–Au–CN-1 composite system) were also circled in Fig. 2d. The interplanar spacing of 0.38 nm shown in the inset in Fig. 2b co-incides with the (010) lattice of WO<sub>3</sub>.

Moreover, the HADDF image and the elemental mapping (for N, C, Au, and W) of sample W-Au-CN-1 are shown in Fig. 2e to i, confirming successful incorporation of Au and WO<sub>x</sub> in the heterostructure system. The TEM images of sample W-Au-CN-2 (with higher W source loading) were also collected for comparison (Fig. 2j, k, and 2m). Typical WO<sub>x</sub> nanobelts are clearly observed in Fig. 2j and k. Compared with sample W-Au-CN-1, the amount of WO<sub>x</sub> incorporated in W-Au-CN-2 increased, due to the fact that the amount of  $WO_x$  nanobelts observed on the surface of the sample W-Au-CN-2 (Fig. 2j) was much higher than that observed on W-Au-CN-1 (Fig. 2b). Well-developed lattice fringes with a 0.38 nm spacing corresponding to the (010) facet of WO<sub>3</sub> were also observed on sample W-Au-CN-2 as shown in Fig. 2m. As the amount of W sources added into the composite system increases, the amount of WO<sub>x</sub> nanobelts grown along the surface of Au-g-C<sub>3</sub>N<sub>4</sub> nanosheets also increases.

To analyze the composition of the WO<sub>x</sub> and Au modified  $g-C_3N_4$  based heterostructure samples, XPS measurements were performed on sample W–Au–CN-1 with the lowest amount of Au and WO<sub>x</sub> incorporated among all the WO<sub>x</sub> and Au modified heterostructure samples, and the XPS results are illustrated in Fig. 3. The C1s spectrum with the C–C (284.8 eV), C–OH (286.4 eV), O–C=O (288.9 eV), C–N/C=O (288.2 eV) components and a satellite peak at 293.5 eV are shown in Fig. 3a. Peaks at 298.5, 399.4 and 400.9 eV in the N 1s spectrum in Fig. 3b were assigned to pyridinic-N, pyrrolic N and graphitic N respectively, among which the pyridinic N and



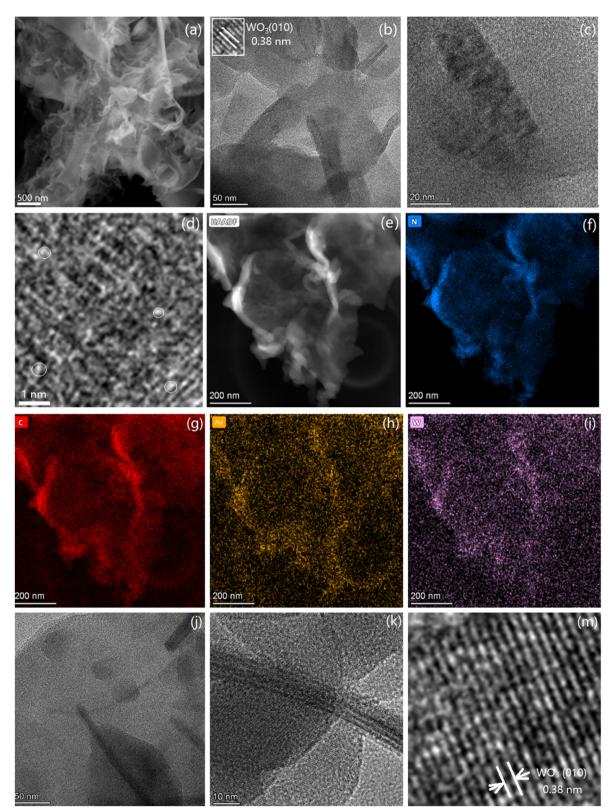
Scheme 1. Construction process of  $WO_x$  and Au modified g- $C_3N_4$  heterostructures using the idea of nanoarchitectonics.



**Fig. 1.** (a) SEM image of sample CN (g-C<sub>3</sub>N<sub>4</sub>). (b to f) TEM images of sample Au–CN-2 at different magnification. (g to j) HADDF image and elemental mapping (N, C, Au) of sample Au–CN-2. (k, m, n) TEM images of sample Au–CN-3 at different magnification. The Au clusters incorporated into the composite system are circled in (d) to (f).

graphitic N were the two main N components and also the key factors that increased the number of active sites and promoted the nucleation of WO<sub>x</sub> nanocomposites in the heterostructure sample. The W4f spectrum in Fig. 3c was fitted into five peaks. The two relatively small peaks at 33.9 and 36.1 eV are attributed to the W4f<sub>7/2</sub> and W4f<sub>5/2</sub> peaks of W<sup>5+</sup> state, respectively, whereas the two large peaks at 35.3 and 37.6 eV correspond to that of the W<sup>6+</sup> state [41]. By creating more oxygen vacancy defects in the WO<sub>x</sub> modified Au-g-C<sub>3</sub>N<sub>4</sub> layered heterostructure system (confirmed by the

observation of  $W^{5+}$  and  $W^{6+}$  states in the W4f spectrum), the charge carrier separation efficiency and thus the photocatalytic performances of the WO<sub>x</sub> modified heterostructure catalysts can be enhanced. The O1s spectrum of sample W–Au–CN-1 in Fig. 3d was fitted into three peaks with oxide (at 530.1 eV with the highest peak intensity) originated from WO<sub>3</sub> in the composite system to be the main component [42]. The peak at 531.5 eV corresponds to the hydroxide species in the composite system while the peak at 532.8 eV can be attributed to the adsorbed organics/H<sub>2</sub>O species in



**Fig. 2.** (a) SEM image of sample W–Au–CN-1. (b to d) TEM image of sample W–Au–CN-1 at different magnification. (e to i) HADDF image and elemental mapping (N, C, Au, W) for sample W–Au–CN-1 with Au clusters circled in (d). (j, k) TEM images of sample W–Au–CN-2. (m) And the inset in (b) show a lattice spacing of 0.38 nm corresponding to the (010) facet of WO<sub>3</sub>.

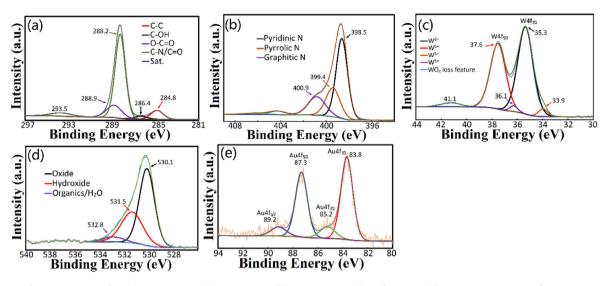


Fig. 3. XPS spectra of sample W-Au-CN-1: (a) C 1s spectrum; (b) N 1s spectrum; (c) W 4f spectrum; (d) O 1s spectrum; (e) Au 4f spectrum.

the sample. Fig. 3e shows the Au 4f spectra of the heterostructure composites W–Au–CN-1. The two peaks located at 83.8 and 85.2 eV correspond to the Au  $4f_{5/2}$  peaks while the peaks at 87.3 and 89.2 eV are related to the Au  $4f_{7/2}$  peaks [43].

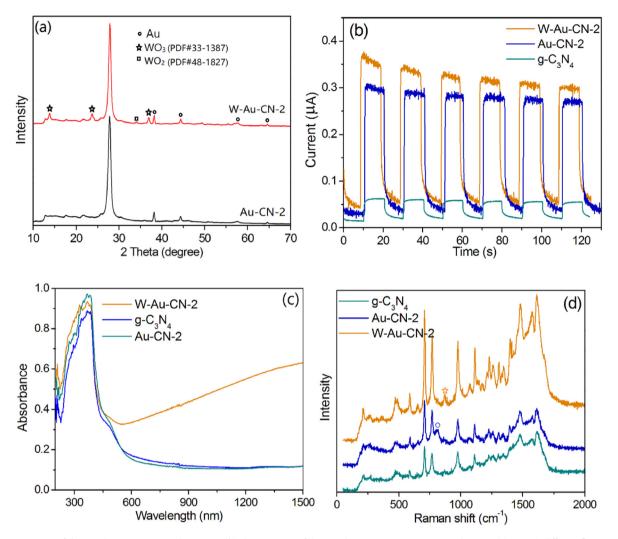
To further investigate the composition of the composite samples, XRD measurements were performed on Au modified g-C<sub>3</sub>N<sub>4</sub> (sample Au-CN-2) and the WO<sub>x</sub> incorporated Au-CN-2 (sample W-Au-CN-2) as illustrated in Fig. 4a. Both Au-CN-2 and W–Au–CN-2 showed diffraction peaks corresponding to the (100) and (002) facets of  $g-C_3N_4$  and the characterisitic peaks of Au, which also confirms the successful incorporation of Au into the heterostructure system. The relatively low intensity of Au peaks observed in the XRD pattern of sample Au-CN-2 and W-Au-CN-2 can be attributed to the fact that the amount of Au introduced into the composites system was scarce. The peak intensity of g-C<sub>3</sub>N<sub>4</sub> related peaks remained almost unchanged after WO<sub>x</sub> modification, which indicates that the primary structure of g-C<sub>3</sub>N<sub>4</sub> nanosheets was maintained after WO<sub>x</sub> immobilization. Moreover, three WO<sub>3</sub> phase peaks and one WO<sub>2</sub> phase peak were observed on the XRD pattern of sample W-Au-CN-2 in Fig. 4a, suggesting that WO<sub>3</sub> was the main WO<sub>x</sub> species in sample W-Au-CN-2 and that only limited amount of WO<sub>2</sub> was formed in the composite system. These results also matched with the XPS data obtained. The presence of both WO<sub>3</sub> and WO<sub>2</sub> phases in the W-Au-CN-2 composite system confirms the formation of surface oxygen vacancies in W-Au-CN-2 heterostructure system, with the addition of reducing agents during the solvothermal treatment process been the key for defect engineering in this case.

Photocurrent and UV–Vis diffuse reflectance measurements were also performed on sample Au–CN-2 and W–Au–CN-2, and the data are shown in Fig. 4b and c. It was obvious that the photocurrent intensity of pristine g-C<sub>3</sub>N<sub>4</sub> nanosheets increased drastically after Au cluster incorporation (as illustrated in Fig. 4b) because of the drastically enhanced conductivity in the presence of Au in the composite system. And a further increase in photocurrent intensity was also observed after WO<sub>x</sub> nanobelts immobilization on Au–CN-2, indicating the successful formation of W–Au–CN-2 heterostructures. In case of UV–Vis diffuse reflectance spectra data shown in Fig. 4c, the diffuse reflectance spectrum of the composite sample (Au–CN-2) remained almost unchanged after Au cluster incorporation, however, a drastic change in the spectrum was observed for the WO<sub>x</sub> modified Au–CN-2 heterostructures (sample

W–Au–CN-2). This can be attributed to the small particle size of the Au clusters. These incorporated Au clusters show enhanced absorption band near blue light region in this case. The extension in light absorption (to the visible/near infrared and full solar spectrum) range, band-to-band transitions, as well as band gap narrowing effects observed for the heterostructure sample W–Au–CN-2 were the keys in improving photocatalytic activity of the heterostructure catalysts [44].

UV-excited Raman measurements were done on pristine g-C<sub>3</sub>N<sub>4</sub>, Au–CN-2, and W–Au–CN-2 sample with the results shown in Fig. 4d. The pristine g-C<sub>3</sub>N<sub>4</sub> nanosheets sample showed a typical g-C<sub>3</sub>N<sub>4</sub> Raman spectrum with relatively lower intensity. An increase in peak intensity was observed for the Au clusters modified g-C<sub>3</sub>N<sub>4</sub> (sample Au-CN-2), indicating the successful embedment of Au during the thermal polymerization treatment stage. The mechanochemical pre-treatment done on the Au source and g-C<sub>3</sub>N<sub>4</sub> nanosheets base before thermal polymerization process was also the key for achieving successful incorporation of Au ions into the g-C<sub>3</sub>N<sub>4</sub> framework forming a highly ordered structure. The WO<sub>x</sub> immobilized sample (W-Au-CN-2) has similar peak positions as that of sample Au–CN-2, but higher peak intensity. This can be attributed to the presence of highly ordered structure of the W-Au-CN-2 heterostructure (maintained even after WOx modification) and also due to the interplay among the different components ( $g-C_3N_4$ , Cu. and  $WO_x$ ) in the composite system. Moreover, the additional peaks observed for the Au and Au/WOx modified composite sample (highlighted using asterisk for sample W-Au-CN-2 and using a circle for sample Au–CN-2 in Fig. 4d) also confirmed the interplay in-between g-C<sub>3</sub>N<sub>4</sub>, Au, and WO<sub>x</sub>.

To evaluate the photocatalytic performances of the heterostructure samples, photocatalytic hydrogen generation and CO<sub>2</sub> photo-reduction measurements (under full solar-spectrum condition, without using any co-catalysts) were performed on the catalyst samples, and the results are illustrated in Table 2. Sample Au–CN-2 showed the highest H<sub>2</sub> generation rate (560 µmolg<sup>-1</sup>h<sup>-1</sup>) among the three Au modified g-C<sub>3</sub>N<sub>4</sub> samples (Au–CN). And sample W–Au–CN-2 with optimized Au and W source ratio revealed the highest H<sub>2</sub> evolution activity among all the seven samples listed in Table 2, which is about 10 times of that of sample Au–CN-2. The H<sub>2</sub> generation rate measured for the WO<sub>x</sub> immobilized sample W–Au–CN-1, W–Au–CN-2, and W–Au–CN-3 were 3990, 5659, and 4100 µmolg<sup>-1</sup>h<sup>-1</sup>, respectively. A significant improvement in



**Fig. 4.** (a) XRD patterns of the sample W–Au–CN-2 and Au–CN-2. (b) Photo currents of the sample W–Au–CN-2, Au–CN-2 and g-C<sub>3</sub>N<sub>4</sub>. (c), UV–Vis diffuse reflectance spectra of the samples. (d) Raman spectra of the samples.

2

Table 2H2, CO, and CH4 evolution performances of samples.

Sample	$H_2$ (µmol/g/h)	CO (µmol/g/h)	CH <sub>4</sub> (µmol/g/h)
g-C <sub>3</sub> N <sub>4</sub>	210	n/a	n/a
Au-CN-1	420	0.48	0.30
Au-CN-2	560	0.89	0.56
Au-CN-3	480	0.63	0.42
W-Au-CN-1	3990	3.62	1.78
W-Au-CN-2	5659	5.64	2.58
W-Au-CN-3	4100	4.21	1.32

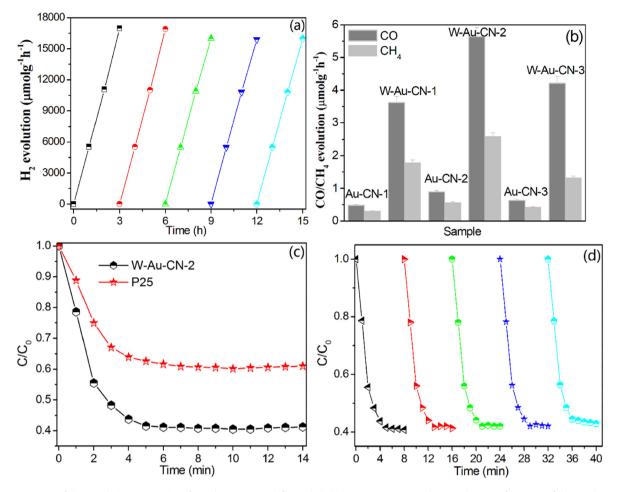
 $H_2$  evolution was observed for the WO<sub>x</sub> and Au modified g-C<sub>3</sub>N<sub>4</sub> heterostructure catalysts in comparison with that of Au (only) modified g-C<sub>3</sub>N<sub>4</sub> samples. In case of photocatalytic H<sub>2</sub> evolution without using any co-catalyst, the activity of these W–Au–CN catalysts can be considered to be reasonably high compared with other similar literature reported ones. Moreover, sample W–Au–CN-2 also showed quite high H<sub>2</sub> generation stability with no obvious decrease in H<sub>2</sub> evolution observed after 5 cycles of testing (as shown on the cyclic stability test plots in Fig. 5a).

In general, in order for photocatalytic water splitting reactions to occur, the band gap of the heterostructure photocatalyst has to be larger than minimum Gibbs free energy required for water splitting (that is 1.23 V in this case). A valence band maximum that is more positive than the oxygen evolution potential (0.82 V vs NHE at pH = 7, equation (1)), and a conduction band minimum that is more negative than the hydrogen generation potential (-0.41 V vs. NHE, at pH = 7, equation (2)) are required for the heterostructure photocatalyst [45–47].

$$H_2O + 2h^+ \rightarrow O_2 + 2H^+ + 4e^-$$
 (1)

$$4H^+ + 4e^- \rightarrow 2H_2 \tag{2}$$

The reason that loading of metal co-catalysts is required during photocatalytic measurements is that co-catalysts are often used as photogenerated charge carrier trap sites enhancing charge separation. However, in the case of  $WO_x/Au$ -g-C<sub>3</sub>N<sub>4</sub> heterostructure composites, no co-catalyst is necessary since Au clusters work as the charge carrier trap sites facilitating H<sub>2</sub> evolution. And the enhanced H<sub>2</sub> generation can be attributed to the formation of these heterostructures in this case. With the  $WO_x$  and Au clusters modified g-C<sub>3</sub>N<sub>4</sub> heterostructures having more positive valence band maximum and more negative conduction band minimum, the redox capability for a variety of reactions can be enhanced. In addition, the immobilization of  $WO_x$  was also able to improve the



**Fig. 5.** (a) Time courses of photocatalytic H<sub>2</sub> generation of sample W–Au–CN-2 (five cycles). (b) Comparison on CO and CH<sub>4</sub> evolution performances of the samples under xenon lamp irradiation (full spectrum condition) with no co-catalyst addition. (c) Photocatalytic NO removal rate comparison. (d) Cyclic NO oxidation stability test for sample W–Au–CN-2.

light absorption capability of the heterostructure composites by extending the light-response range of the material to the near infrared region. The detailed photocatalytic mechanism of these heterostructures will be discussed later.

To further investigate the photocatalytic activity of the WO<sub>x</sub>/Aug-C<sub>3</sub>N<sub>4</sub> heterostructures, CO<sub>2</sub> photoreduction test was done on the samples and the data are listed in Table 2. CO<sub>2</sub> photoreduction for carbon-based fuel (e.g., methane, carbon monoxide, methanol etc.) production has been hot topics recent years. In order to have the carbon-based fuel evolution reactions (from CO<sub>2</sub> photoreduction) to occur, a more negative conduction band minimum in comparison with that of the standard carbon-based fuel (e.g., CO and CH<sub>4</sub>) evolution potential (in V vs. NHE at pH = 7, shown in *eq.* 3 and 4) is often required for the catalysts [48,49].

 $CO_2 + 8 H^+ + 8e^- \rightarrow CH_4 + 2H_2O E^o = -0.24 V$  (3)

$$CO_2 + 2 H^+ + 2e^- \rightarrow CO + H_2O E^0 = -0.52 V$$
 (4)

Fig. 5b shows the histogram chart for the comparison of CO and CH<sub>4</sub> evolution rate in full solar spectrum irradiation condition of the Au clusters and WO<sub>x</sub>/Au immobilized composite samples and with no co-catalyst addition. In comparison with the extremely poor CO<sub>2</sub> photoreduction performance of pristine g-C<sub>3</sub>N<sub>4</sub> nanosheets, a significant increase in CO and CH<sub>4</sub> evolution was observed for the Au clusters modified g-C<sub>3</sub>N<sub>4</sub> composites (Au–CN). This is due to the

fact that by introducing of Au clusters and WO<sub>x</sub> to the g-C<sub>3</sub>N<sub>4</sub> network, bandgap narrowing accompanied by the adjustment on the band edge position occurs, leading to extending in light absorption range of the heterostructures. Sample W–Au–CN-2 revealed the best CO and CH<sub>4</sub> evolution performance among all the heterostructure samples with a CO generation rate of 5.64  $\mu$ molg<sup>-1</sup>h<sup>-1</sup> and a CH<sub>4</sub> evolution rate of 2.58  $\mu$ molg<sup>-1</sup>h<sup>-1</sup>. Similar trend was also observed for the photocatalytic H<sub>2</sub> evolution data. The heterostructure nanoarchitectonics of WO<sub>x</sub>/Au-g-C<sub>3</sub>N<sub>4</sub> is the key in attaining efficient CO<sub>2</sub> photoreduction.

Photocatalytic NO removal test was also performed on the heterostructure samples. NO removal occurs when the photogenerated electrons migrate to the valence band of photocatalyst and react with the  $O_2$  adsorbed on the surface of the material forming  $\cdot O^{2-}$  species which can be then used for oxidizing NO into  $NO_3^-$ . The detailed reactions are shown as follows (equations (5) and (6))

$$e_{CB}^- + O_2 \to {}^{\bullet}O_2^- \tag{5}$$

$$NO + {}^{\bullet}O_2^- \rightarrow NO_3^- \tag{6}$$

Fig. 5c shows the NO removal plots obtained for sample W–Au–CN-2 using the NO removal performance data of P25 for comparison. Sample W–Au–CN-2 was able to remove about 61% of the NO within 6 min while P25 was able to remove only 40%. The

NO removal cyclic stability test results obtained for sample W-Au-CN-2 (in Fig. 5d) shows that the NO removal efficiency of sample W-Au-CN-2 remained almost unchanged after 5 cycles. The constructed WO<sub>x</sub>/Au-g-C<sub>3</sub>N<sub>4</sub> heterostructures can be considered a highly efficient semiconducting photocatalyst that can be used for many photocatalytic applications due to the narrowed bandgap, abundant surface oxygen vacancies (due to the presence of  $WO_3$  and  $WO_2$  in the  $WO_x$  component of the heterostructure), extended light absorption range (due to the intense localized surface plasmon resonance absorption of WO<sub>x</sub> in both visible and near infrared region) and thus enhanced photocatalysis [50]. The abundant surface oxygen vacancies present on the heterostructures is able to cause collective oscillations of excess charges (electrons) that is similar to that of the localized surface plasmon resonance effect of noble metals, leading to improved photocatalysis. In order to investigate the active species during photocatalytic reactions, quenching experiments for rhodamine B (RhB) photodegradation were performed using WOx/Au-g-C3N4 heterostructures (sample W-Au-CN-2) as the catalyst and AgNO<sub>3</sub>, disodium salt dihydrate (EDTA-Na<sub>2</sub>), benzoquinone (BQ), isopropanol (IPA) as the scavenger to capture  $e^-$ ,  $h^+$ ,  $\bullet O^{2-}$ ,  $\bullet OH$ , respectively. It was found that  $\bullet O^{2-}$ , produced in the WO<sub>x</sub>/Au-g-C<sub>3</sub>N<sub>4</sub> heterostructures system, was the main active species for RhB photodegradation reations. And both •O<sup>2-</sup> and •OH contribute significantly to improving the photocatalytic activities of the WO<sub>x</sub>/Au/g-C<sub>3</sub>N<sub>4</sub> heterostructures system.

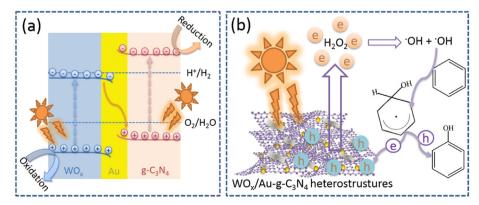
According to the results obtained, charge transport in WO<sub>x</sub>/Aug-C<sub>3</sub>N<sub>4</sub> photocatalyst does not follow the type II mechanism, but a direct solid-state Z-scheme mechanism (as proposed in Scheme 2a) with different charge transfer paths that facilitates charge carrier separation/transfer and enhances photocatalytic performances of the catalyst. There exist three charge transport channels in the WO<sub>x</sub>/Au-g-C<sub>3</sub>N<sub>4</sub> Z-scheme photocatalytic system involving: electron excitation from valence band to conduction band; electron excitation from valence band to the oxygen vacancies below the conduction band; and electron excitation caused by localized surface plasmon resonance. Au cluster incorporation is the key in switching the interfacial charge transfer routes of g-C<sub>3</sub>N<sub>4</sub> and WO<sub>x</sub> from type II to direct Z-scheme with electrons in the conduction band of WO<sub>x</sub> transported to the valence band of g-C<sub>3</sub>N<sub>4</sub>. In this case, Au cluster works as the electron sink that can assist capture and transport of the photogenerated electrons, facilitating charge separation and therefore improving the photocatalytic activities.

Benzene oxidation measurement was also performed on the pure  $g-C_3N_4$  nanosheets sample and W-Au-g-CN-2 composites. The pure  $g-C_3N_4$  nanosheets sample could not catalyse benzene-to-phenol oxidation reaction because of its band gap limitations. According to literature, the incorporation of noble metals is often

required in this case to facilitate charge transfer from the metallic part of the composite catalyst to the lowest unoccupied molecular orbital of the g-C<sub>3</sub>N<sub>4</sub> part in order to catalyse benzene oxidation reactions with the assistant of H<sub>2</sub>O<sub>2</sub> [34]. Sample Au-g-CN-2 showed a phenol yield of only 5% in the case of using acetonitrile as the solvent but with no H<sub>2</sub>O<sub>2</sub> addition. H<sub>2</sub>O<sub>2</sub> can be considered to be one of the key components during the benzene-to-phenol photooxidation reactions since the photo-excited electrons in the conduction band of the catalyst have to be transferred to H<sub>2</sub>O<sub>2</sub> to generate hydroxyl radicals that can be used for further oxidation reactions [47]. Under dark condition, the heterostructure sample W-Au-g-CN-2 showed 43.5% of benzene-to-phenol conversion rate and a phenol selectivity of 52.1%. A drastic enhancement in benzene-to-phenol conversion efficiency and selectivity (to phenol) was observed for sample W-Au-g-CN-2 under full solar spectrum light irradiation condition. The selectivity and the conversion rate measured for sample W-Au-CN-2 reached 99.3% and 88.1%, respectively, suggesting the existence of synergistic effects between the Au clusters, WO<sub>x</sub> nanocomponents and ultrathin g-C<sub>3</sub>N<sub>4</sub> in the heterostructure system. The schematic illustration of the proposed mechanism for benzene photo-oxidation reactions using  $WO_x/Au-g-C_3N_4$  heterostructure photocatalyst under full solar spectrum irradiation condition is shown in Scheme 2b with hydroxyl radicals to be the key active species. The incorporation of homogeneously distributed Au clusters is necessary due to the fact that it can facilitate hydroxyl radical generation in the g-C<sub>3</sub>N<sub>4</sub>-based composite catalyst system, promote charge transport in-between the g-C<sub>3</sub>N<sub>4</sub>/WO<sub>x</sub> phases and form hydroxy cyclohexadienvl radicals (via reaction between hydroxyl radical and benzene) that works as the key intermediates in initiating further oxidation reaction by reacting with the photogenerated holes for attaining phenol production [48,51].

# 4. Conclusion

To have control over the thermal polymerization reactions, a mechano-chemical pre-treatment process was applied to incorporate Au clusters into g-C<sub>3</sub>N<sub>4</sub> nanosheets base. Small Au NPs and clusters were homogeneously embedded in ultrathin g-C<sub>3</sub>N<sub>4</sub>. The Au modified g-C<sub>3</sub>N<sub>4</sub> nanosheets showed enhanced photocurrent generation which also in consistency with the photocatalytic performances results (H<sub>2</sub> generation and CO<sub>2</sub> reduction). A solvothermal synthesis method was applied to incorporate WO<sub>x</sub> nanobelts on the Au modified g-C<sub>3</sub>N<sub>4</sub>. WO<sub>x</sub> nanobelts with abundant oxygen vacancies were decorated on Au-modified g-C<sub>3</sub>N<sub>4</sub> nanosheets to form the two dimensional (2D)/2D WO<sub>x</sub>/Au-g-C<sub>3</sub>N<sub>4</sub> heterojunctions having excellent  $CO_2$ photo-reduction



Scheme 2. (a) Schematic diagram of electron transport in Z-scheme photocatalytic system. (b) Proposed mechanism for catalytic benzene oxidation using WO<sub>x</sub>-Cu-g-C<sub>3</sub>N<sub>4</sub> heterojunctions under light irradiation.

performances in full solar spectrum condition (with a CO and CH<sub>4</sub> generation rate of 5.64 and 2.58  $\mu$ molg<sup>-1</sup>h<sup>-1</sup>, respectively). The presence of Au clusters in the heterostructure system (constructed using the idea of nanoarchitectonics) is the key that leads to the double Z-scheme charge transport system formation with improved charge separation/transport efficiency due to the builtup direct Z-scheme charge transfer between Au and  $g-C_3N_4$  (or  $WO_{v}$ ) as well as the indirect charge transfer between  $WO_{v}$  and g-C<sub>3</sub>N<sub>4</sub>. The NO removal rate of the WO<sub>x</sub>/Au-g-C<sub>3</sub>N<sub>4</sub> heterojunctions with optimized composition ratio reached 61%. This work may contribute to offer novel strategies for the design of hierarchical g-C<sub>3</sub>N<sub>4</sub> hybrid nanostructures that can be used as promising photocatalysts for solar energy conversion and pollutant treatment. The constructed WO<sub>x</sub>/Au-g-C<sub>3</sub>N<sub>4</sub> heterojunctions catalyst showed outstanding benzene to phenol conversion rate (88.1%) and excellent selectivity (99.3%) under full solar spectrum irradiation as well as high stability and cyclability, possessing huge potential in direct conversion of benzene to phenol.

#### **Credit author statement**

Xiao Zhang: Conceptualization, Methodology, Investigation, Writing – original draft, Writing – review & editing. Katarzyna Matras-Postolek: Writing – review & editing. Ping Yang: Investigation, Writing – review & editing.

### **Declaration of competing interest**

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

#### Data availability

Data will be made available on request.

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